

# Nano/Energy Journal Club

## **Topic: Intermediate Band Solar Cells**

Led by: Kyle Montgomery

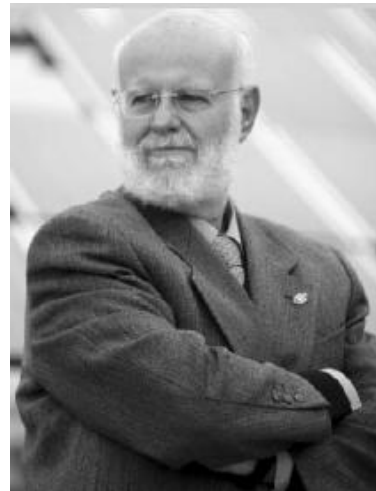
01/15/2010

# Intermediate Band Solar Cell (IBSC)

- Paper: A. Luque and A. Martí, "The Intermediate Band Solar Cell: Progress Toward the Realization of an Attractive Concept," *Advanced Materials*, vol. 22, pp. 160-174, 2010.
- Paper received: July 17, 2009
- These guys first described the concept in Phys. Rev. Lett. in 1997.

Luque and A. Martí, "Increasing the Efficiency of Ideal Solar Cells by Photon Induced Transitions at Intermediate Levels," *Physical Review Letters*, vol. 78, p. 5014, 1997.

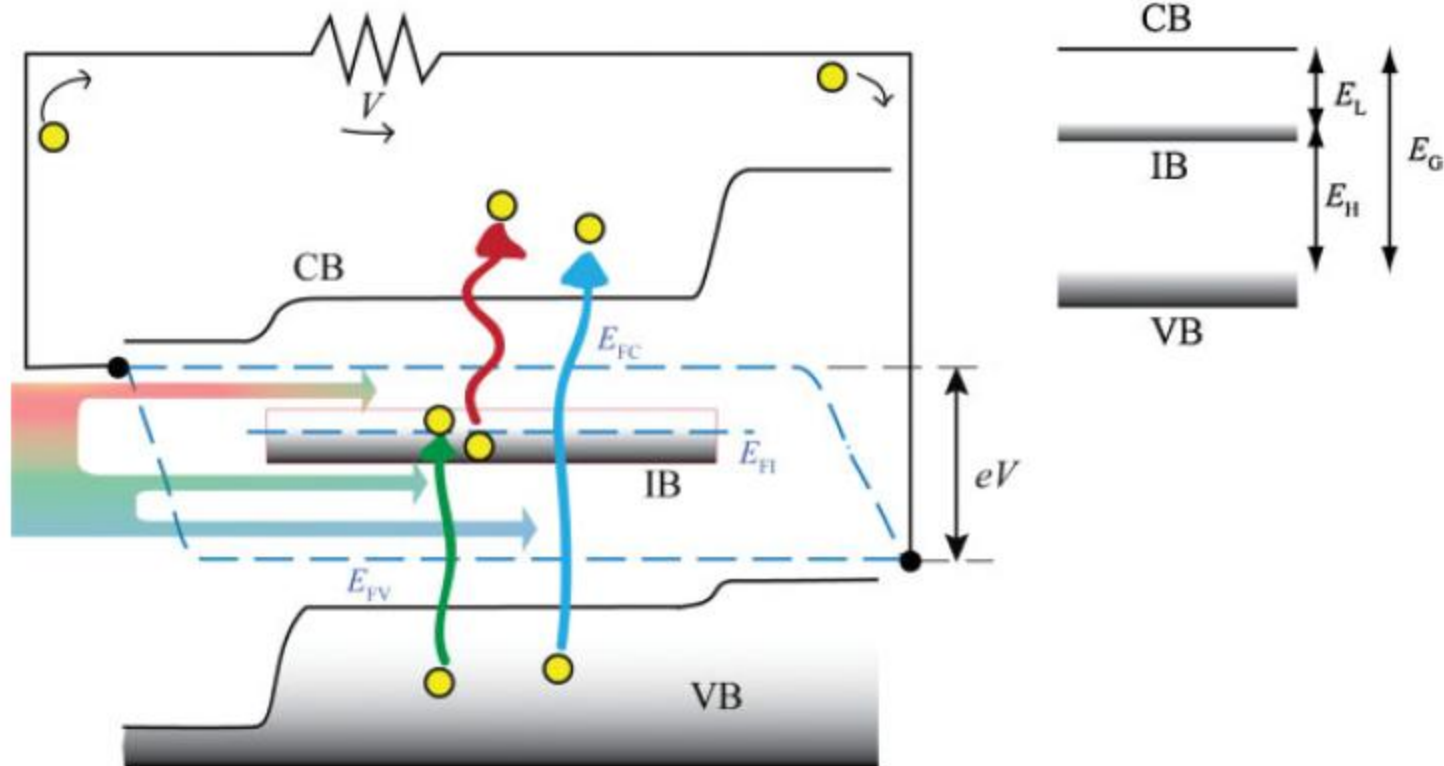
Antonio Luque



Antonio Martí



# IBSC: The Basics



# IBSC: Benefits

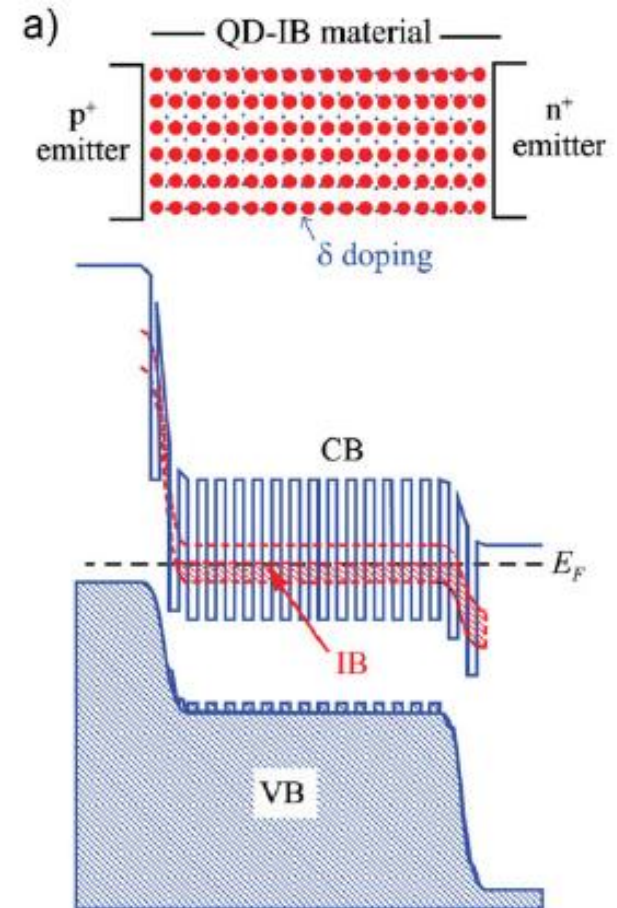
- Increased photocurrent due to absorption of sub-bandgap photons
- No reduction in voltage (ideally) over single bandgap cell
- Limiting efficiency: 63.2% for IBSC vs. 40.7% for single junction (assuming max concentration)

# IBSC: Essentials

- 3 quasi-Fermi levels (CB, VB, IB)
- IB is disconnected from n- and p-contacts so that extracted voltage is the difference between CB and VB
- Selective absorption
- IB needs to be partially filled with electrons
- OPTION: Energy transference - 2 sub-bandgap photons pump 2 electrons from VB to IB, and then 1 electron returns to VB and pumps the other to the CB

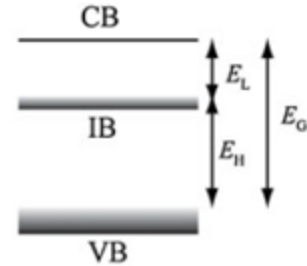
# Minibands through Quantum Dots

- QDs (as opposed to wells or wires) are necessary to isolate the IB from the CB through zero DOS
- QDs typically undoped, so the barrier layer in between is n-type



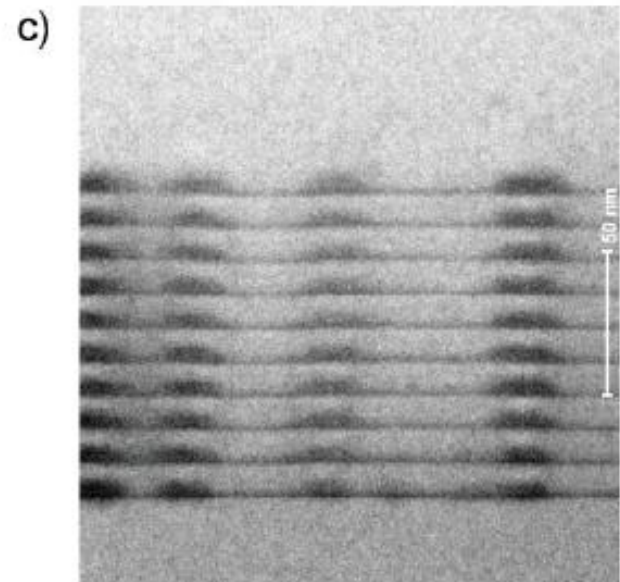
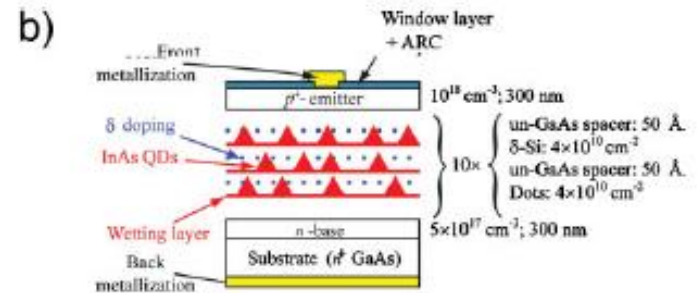
# Optimal QD Design

- Optimal bandgaps: 0.71, 1.24, 1.95eV
- Corresponds to  $\text{In}_{0.55}\text{Ga}_{0.45}\text{As}$  QDs in  $\text{Al}_{0.44}\text{Ga}_{0.56}\text{As}$  barrier material after strain is considered
- Apparently, this is non-trivial in terms of growth, so InAs QDs in GaAs barrier has been studied (0.30, 1.02, and 1.30eV)



# InAs QDs in GaAs

- Grown by MBE under the Stranski-Krastanov growth mode
- S-K growth means a wetting layer is grown first, which introduces a quantum well
- QDs not spherical

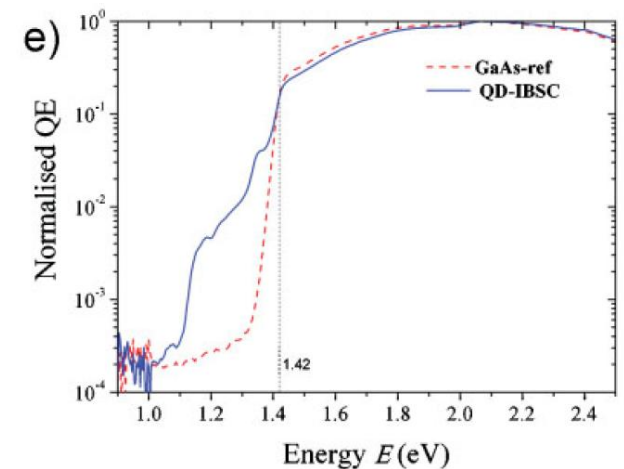
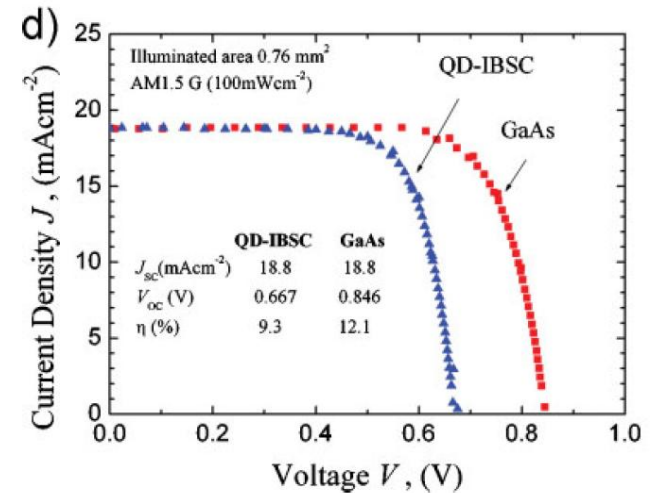


[35] A. Marti, N. Lopez, E. Antolin, E. Canovas, A. Luque, C. R. Stanley, C. D. Farmer, P. Diaz, *Appl. Phys. Lett.* **2007**, *90*, 233510.



# I-V and EQE Results

- Though I-V shows minimal increase in current, EQE clearly shows boost in long wavelength absorption
- Loss of voltage leads to lower efficiency compared to single junction GaAs



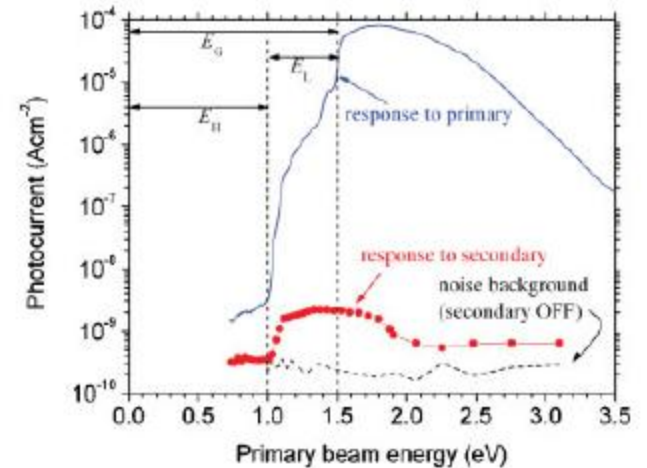
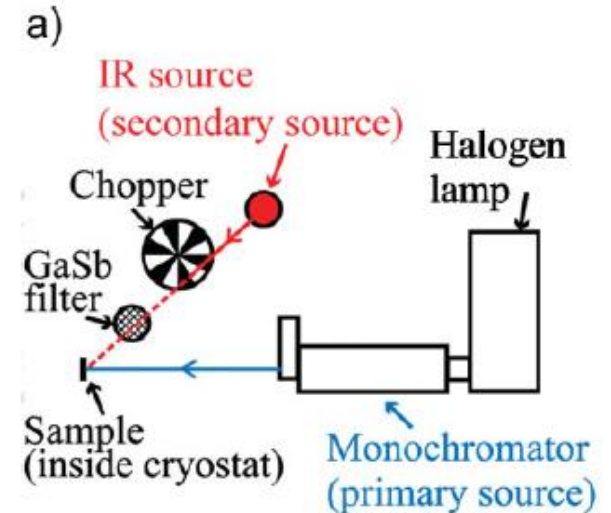
[36] A. Luque, A. Martí, N. López, E. Antolín, E. Cánovas, C. R. Stanley, C. Farmer, P. Díaz, *J. Appl. Phys.* 2006, 99, 094503.

# Thermodynamics

- Thermodynamic requirement: absorption of a second photon is necessary to getting full voltage (between CB and VB) out.

# Proving Two-Photon Absorption

- IB  $\rightarrow$  CB transition was thought to be due to thermal escape or field-assisted tunneling, so the following experiment was performed
- IR source provided boost in current

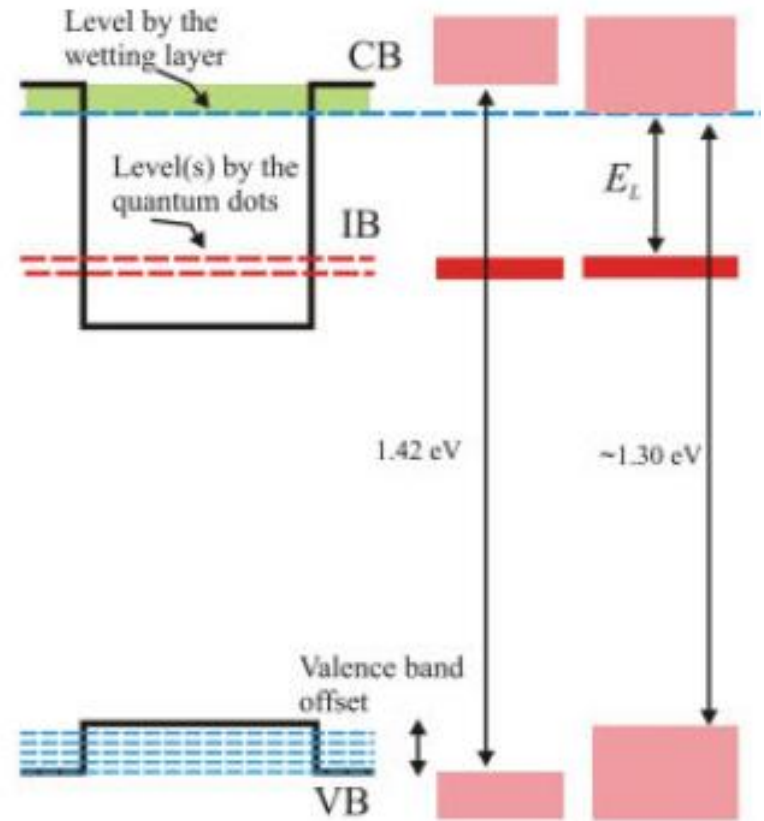


# Improving Current

- Increase layers of QDs
  - Result: sub-bandgap current is great, but overall photocurrent is terrible due to dislocations in emitter
- Add stress-compensating layers (GaP, GaNAs, etc)
  - Result: Improves current slightly, but voltage is reduced due to lowering of effective bandgap

# Improving Voltage

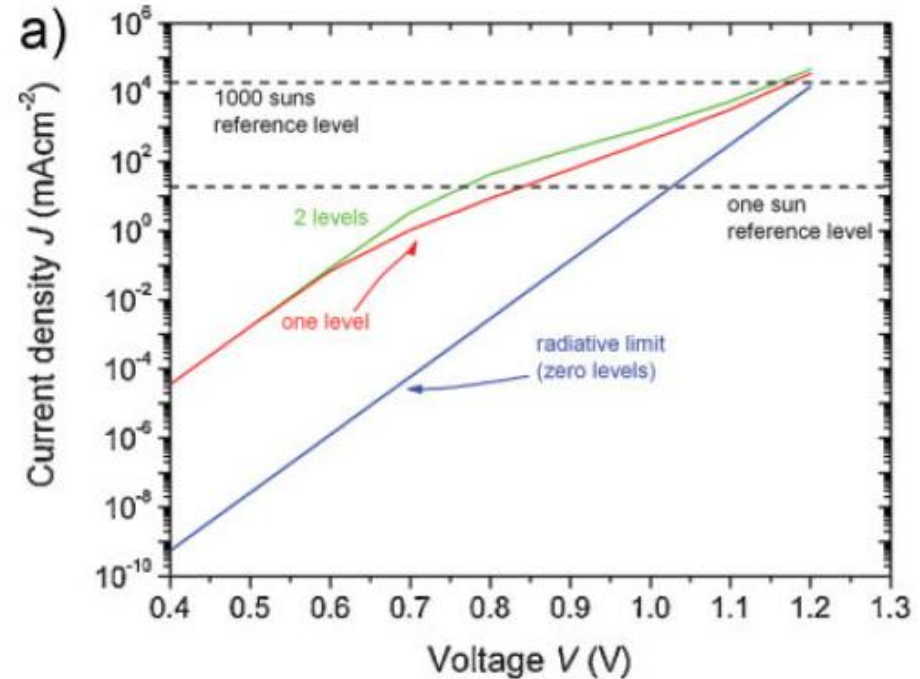
- Reduction possibly due to lower effective bandgap in the nanostructured material
  - compensate by using higher bandgap barrier material (AlGaAs)



[10] A. Marti, E. Antolin, E. Canovas, N. Lopez, P. G. Linares, A. Luque, C. R. Stanley, C. D. Farmer, *Thin Solid Films* 2008, 516, 6716.

# Improving Voltage (cont.)

- QDs actually result in multiple energy levels in the bandgap (perhaps dependent on shape/size)
  - more recombination paths ( $V_{oc} \propto \ln(J_L/J_0)$ )
  - lower voltage



**Dark I-V for 1.3eV IBSC**

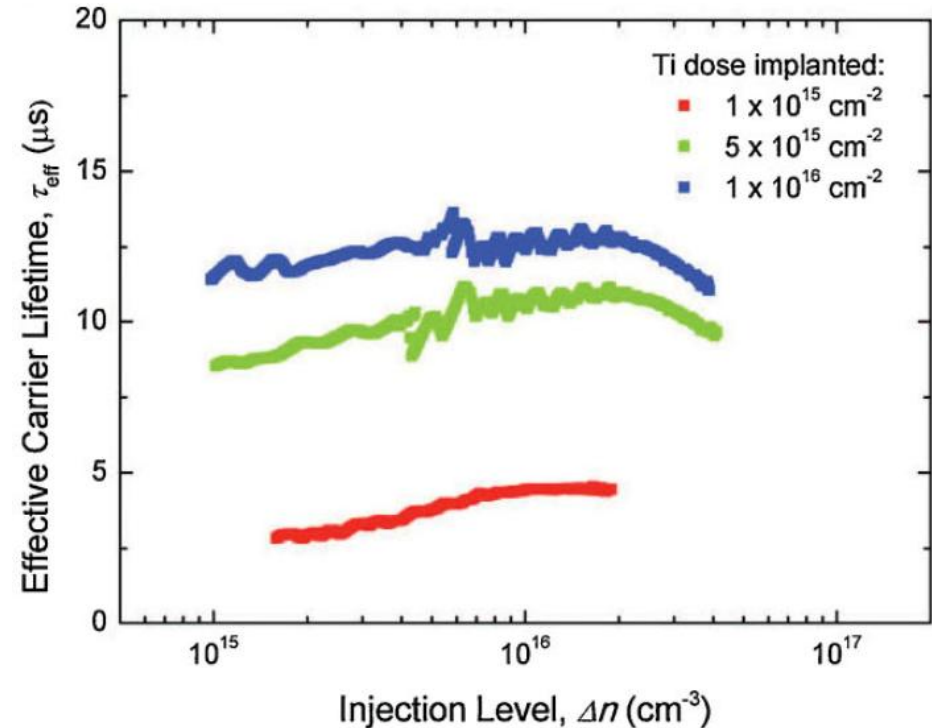
[10] A. Marti, E. Antolin, E. Canovas, N. Lopez, P. G. Linares, A. Luque, C. R. Stanley, C. D. Farmer, *Thin Solid Films* 2008, 516, 6716.

# Bulk IB Materials

- Benefit: higher density of band states over QD-based devices (limited to  $\sim 10^{17}/\text{cm}^3$ ) leading to higher sub-bandgap absorption
- Requirement: sufficiently deep level to prevent thermal transfer from IB to CB ( $> 0.2\text{eV}$ )

# Impurity Bands

- SRH recombination through deep levels is prevented by reaching a critical concentration of impurities
- Due to a delocalization of impurity wave functions
- Titanium in Silicon is experimentally shown



[24] E. Antolin, A. Marti, J. Olea, D. Pastor, G. Gonzalez-Diaz, I. Martil, A. Luque, *Appl. Phys. Lett.* **2009**, 94, 042115.



# Bulk IB Material Options

- Highly Mismatched Alloys (HMAs): III-V or II-VI material with N or O replacing V or VI anions (e.g. ZnTe:O, ZnMnOTe)
  - have demonstrated 100% increase in  $J_{sc}$ , 15% decrease in  $V_{oc}$ , 50% increase in power conversion w.r.t. undoped cell (BUT,  $\eta < 1\%$ )
- Transition Metals (e.g. V in  $In_2S_3$ )
  - difficult to make, no solid results yet
- Impurity bands (e.g. Ti in Si, Fe in  $CuGaS_2$ )

# State of the art IBSC...

- Efficiency: 18.3% (Blokhin et al.)
- Voc: 0.9V, AM0 (Hubbard et al.)
- Current increase over single junction: 23%, 24.26 mA/cm<sup>2</sup>, though doubt remains on the cause of the increase (Okada et al.)

# IBSC: Concluding Remarks

- Clearly, material growth/synthesis is an inhibitor for either type of IBSC.
  - Are there other ways of integrating QDs in a p-n junction?
  - Can the shape and spacing of the QDs be controlled any better in S-K growth?
- Proper characterization is key (EQE by 2 lamp sources). Beware of results published w/out mention of doing this.
- **Fundamental Limitation: Dark current increases due to addition of the IB!!**